Magneto-Optical Response Enhanced by Mie Resonances in Nanoantennas

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Supporting Information

ABSTRACT: We demonstrate both experimentally and numerically multifold enhancement of magneto-optical effects in subwavelength dielectric nanostructures with a magnetic surrounding exhibiting localized magnetic Mie resonances. We employ amorphous silicon nanodisks covered with a thin nickel film and achieve the 5-fold enhancement of the magneto-optical response of the hybrid magnetophotonic array of nanodisks in comparison with a thin nickel film deposited on a flat silica substrate. Our findings allow for a new basis for active and nonreciprocal photonic nanostructures and metadevices, which could be tuned by an external magnetic field.

KEYWORDS: Mie resonances, silicon nanophotonics, subwavelength nanostructures, optical magnetism, magneto-optical effects

Control of light by an external magnetic field is one of the important methods for modulation of its intensity and polarization.1 Magneto-optical effects at the nanoscale are usually observed in magnetophotonic crystals,7−5 nanostructured hybrid materials,6,20 or magnetoplasmonic crystals.10−18 The early studies on the enhancement of magneto-optical response in nanostructured materials were done more than 30 years ago.19−22 An indirect action of an external magnetic field (e.g., through the Faraday effect) is explained by the fact that natural materials exhibit negligible magnetism at optical frequencies. However, the concept of metamaterials overcomes this limitation imposed by nature by designing artificial subwavelength meta-atoms that support a strong magnetic response, usually termed as optical magnetism, even when they are made of nonmagnetic materials.23 The fundamental question is what would be the effect of the interaction between an external magnetic field and an optically induced magnetic response of metamaterial structures.24 Here we make the first experimental step toward answering this fundamental question and demonstrate the multifold enhancement of the magneto-optical response of nanoantenna lattices due to the optical magnetism.

Nanophotonics is often associated with plasmonic structures made of metals such as gold or silver. However, it is known that plasmonic structures suffer from high losses of metals, heating, and incompatibility with CMOS fabrication processes. Recent developments in the nanoscale optical physics gave birth to a new branch of nanophotonics aiming at the manipulation of optically induced Mie-type resonances in dielectric nanoparticles made of materials with high refractive indices.23,25−27 It has been shown recently that resonant dielectric structures offer unique opportunities for reduced dissipative losses and large resonant enhancement of both electric and magnetic fields. High-index dielectric structures can be employed as new building blocks to obtain unique functionalities such as magnetic Fano resonances,28,29 highly transmittable metasurfaces,33,34 and novel metadevices.35,36 Here we extend the concept of high-index resonant nanophotonics to the case of magnetically active materials and study the magneto-optical (MO) response of a dielectric array of nanodisks covered with a thin magnetic film, as shown schematically in Figure 1. We emphasize that at the microscopic level the optical response is driven by electric dipoles, but high-index dielectric nanoparticles with this microscopic response generate effectively magnetic multipoles. Despite its geometrical simplicity, a nanoantenna is a resonant element that supports both electric and magnetic multipolar modes that can generate efficiently optical magnetism effects. We employ such effects here and demonstrate, for the first time to our knowledge, that the optical magnetism associated with the magnetic dipole Mie resonance in a single silicon nanodisk leads to the multifold enhancement of the magneto-optical response of the hybrid magnetophotonic array of nanodisks in comparison with a thin nickel film deposited on a flat silica substrate.

We consider samples composed of a thin nickel film deposited on top of an array of silicon nanodisks with different diameters and study optical and magneto-optical response of such hybrid planar nanostructures. Nickel has been chosen as
magnetic material because it has a high magnetic activity and possesses strong magnetic response, as it has been shown in the recent works.\textsuperscript{30}−\textsuperscript{32} We also examined for comparison dielectric magnetic materials like yttrium iron garnet, and carried out the numerical simulations for the structure, where the 5 nm thick nickel film is replaced by the 5 nm-thick Bi:YIG film (see Supporting Information, "Comparison of magneto-optical responses of the structure in case of garnet and nickel film" section). Our results show that magneto-optical enhancement for the structure with YIG film is 65\times smaller than for the same sample with Ni film. Moreover, fabrication process for samples with YIG may affect the nanostructured a-Si disks, because YIG films must be annealed under high temperatures to become magneto-optically active. In addition with the aim of confirming the high efficiency and low losses of the considering structure we performed numerical calculation of the absorption coefficient for both structure under study and all-dielectric structure without Ni film (see Supporting Information, "Nickel
The Ni film contribution to the absorption of the structure. The 5 nm thick Ni film changes the absorption for only 10–15% in comparison with the structure without Ni film. The shape of the nanodisks is chosen to match the position of the lowest Mie-type resonances. Indeed, it has been shown that the electric and magnetic dipolar resonances can be overlapped spectrally by changing both height and diameter of the nanodisk, leading to a significant increase of the local fields inside a dielectric nanoscale structure.

For experimental studies, we use arrays of hydrogenated amorphous silicon nanodisks placed on a transparent silica substrate. The sample is covered with the 5 nm-thick nickel film in order to achieve the magneto-optical response in an external magnetic field. The SEM image of the sample is shown on Figure 1. The disk diameter \(d\) varies from 320 to 360 nm and the height of the nanodisks \(h\) is fixed, being equal to 220 nm. The magnetic dipole Mie resonance of an isolated nanoantenna is designed to be in the visible and near-IR spectral ranges. The lattice spacing \(a\) of the array is 830 nm, so that the contribution of diffraction is minimal because the phase-matching condition of the first diffraction maximum is satisfied only for the IR part of the spectrum due to the sample geometry. The contribution of higher-order resonances is negligible.

In experiment, we use normal incidence illumination and apply an external magnetic field along the sample surface and perpendicularly to the wave vector (see Figure 1). The incoming polarization is chosen to satisfy the right-hand rule: the electric field is perpendicular to both wave vector and direction of external magnetic field. This configuration is known as the Voigt geometry for transmission measurements.

The experimental geometry is similar to that used for the observation of the transverse magneto-optical Kerr effect (TMOKE), being an intensity-dependent effect observed for reflectance. An external magnetic field changes the refractive index of a magnetic medium. This action leads to a shift of the transmittance spectrum \(\Delta \lambda\) relative to the nonmagnetized case. The MO response of the sample can be defined as following:

\[
\frac{T(H) - T(-H)}{T(0)}
\]

where \(T\) is the transmittance and \(H\) is the applied external magnetic field. If the spectral shift is small (\(\Delta \lambda \ll \lambda\)), the MO response is proportional to the spectral derivative of the transmittance, \(\partial T/\partial \lambda\). It means that the maximum of the MO signal does not coincide with a dip or a peak in the transmission spectrum, but it corresponds to a maximal slope of the resonance curve. TMOKE is odd in the magnetization and could be observed for an oblique incidence: the sign of the effect changes with changing the angle of the incidence \(\theta\) to \(-\theta\) or reversing magnetic field direction. In the case of normal incidence, the MO response becomes weaker because it is proportional to the square of the magnetization. Due to this, the MO signal in the experiment is locked-in to the doubled frequency of an alternating external magnetic field.

In order to characterize optically our MO samples, we perform measurements of the transmittance with the setup described in the Methods section (Experimental Setup). Figure 2 shows the experimental results for one of the samples. The sample parameters are \(a = 830\) nm, \(d = 340\) nm, and \(h = 220\) nm. The dip in the transmission at the wavelength of 685 nm corresponds to the excitation of the magnetic dipole Mie-type resonance in a single silicon nanodisk, which has been
confirmed by numerical simulations (see Figures 3 and 4). The magneto-optical response shows a resonant enhancement in the vicinity of magnetic dipole resonance of the nanodisks (blue curve). The dependence of the magneto-optical effect on the magnitude of the applied external magnetic field is shown in Supporting Information. The magneto-optical spectrum measured for the nickel film does not have any resonant features compared to the sample. Thus, we can easily trace the correlation between the magneto-optical enhancement and the dipole magnetic resonance excitation in the silicon nanodisks.

In order to confirm this idea, we fabricate a set of samples with increasing diameters of the silicon nanodisks. Depending on the disk diameter, the resonant wavelength is shifted: the wider the disk the redder the resonant wavelength. Figure 3 shows the spectral dependencies of the transmittance (a) and the MO response (b) for samples with different diameters of the nanodisks.

We observe the resonance shifts from the wavelength of 680 nm (for \( d = 340 \) nm) to 720 nm (for \( d = 360 \) nm). The corresponding shift of the MO response is illustrated in Figure 3c. To confirm our experimental data, we calculate the transmittance and MO response spectra using the finite-difference time-domain (FDTD) technique in the Numerical FDTD Solutions software (see Methods, Numerical Simulations). In our simulations, the refractive index of glass, the complex dielectric permittivity of Si disks, and the Ni film are all taken into account.

The comparisons between experimental results and numerical simulations are shown in Figure 4. Upper graphs illustrate experimental and numerical results for (a) transmittance and (b) magneto-optical response, for an array of silicon nanodisks with a diameter of \( d = 344 \) nm, a height of \( h = 220 \) nm, and a lattice spacing \( a = 830 \) nm, covered with a 5 nm thick nickel film. The experimental and numerical spectra show quite similar behavior, but they differ in the absolute values. This difference could be caused by the local imperfections in the form and size of nanoparticles and the roughness of the deposited Ni film, which could bring about additional scattering losses and lead to a decrease in the transmission and magneto-optical response. The lower graphs in Figure 4 demonstrate the calculated (c) electric and (d) magnetic fields inside the nanodisk in the vicinity of the dipole magnetic Mie resonance at \( \lambda = 680 \) nm.

Figure 4. Comparison between experimental data and numerical results. Experimental (black dots) and numerical (blue solid lines) transmittance spectra (a) and magneto-optical response spectra (b) for an array of silicon nanodisks with a diameter of \( d = 344 \) nm, a height of \( h = 220 \) nm, and a lattice spacing \( a = 830 \) nm, covered with a 5 nm thick nickel film (c, d) Calculated electric and magnetic fields inside the nanoantenna in the vicinity of the dipole magnetic Mie resonance at \( \lambda = 680 \) nm.
magnetic and electric dipole resonances” section). Numerical results demonstrate that the magneto-optical effect enhancement caused by the magnetic dipole resonance excitation is several times stronger than in the case of the electric dipole resonance. Despite the fact that amplification of the local electric and magnetic fields in the region of magnetic and electric dipole resonances is comparable in magnitude, and also the slope of the resonance curve (see Figure S2 in the Supporting Information) in the vicinity of the electric dipole resonance exceeds the slope for the magnetic dipole resonance, the enhancement of the magneto-optical response is several times larger in the region of magnetic dipole resonance in comparison with the electric one. These results proof that magneto-optical effects are not only driven by the simple local field concentration in the nanoresonators, but also depend on the mode character.

In summary, we offer the first demonstration of the magneto-optical effects enhanced by optically induced magnetic dipole Mie resonances, manifesting a strong interaction between magnetic properties and induced optical magnetism. Our unique findings allows for novel approaches in a magnetic control of recently reported strong nonlinear effects in nanoparticles, as well as the realization of the time-reversal symmetry breaking at the nanoscale for photonic topological insulators.

Our results could create a new basis for novel active and nonreciprocal nanophotonic structures and meta-devices, which could be tuned by external magnetic fields.

■ METHODS

Fabrication of Samples. In the capacity of a base material for our nanodisks, we use a 220 nm-thick hydrogenated amorphous silicon film deposited atop a silica substrate, using plasma-enhanced chemical vapor deposition technique in the Oxford PlasmaLab System 100. The deposition is carried out under the following conditions: 25 sccm SiH₄ and 475 sccm He at 250 °C for approximately 10 min. In order to nanostructure Si films, we use the electron-beam lithography with the subsequent reactive ion etching. For that purpose we spin-coat over the sample the negative tone ma-N-2403 electron beam resist with a thickness of 300 nm and a water-soluble anticharging conductive polymer (ESPACER 300Z) to avoid static charging. Then we expose the resist using an electron-beam anticharging conductive polymer (ESPACER 300Z) to avoid over the sample the negative tone ma-N-2403 electron beam subsequent reactive ion etching. For that purpose we spin-coat devices, which could be tuned by external magnetic active and nonreciprocal nanophotonic structures and meta-devices, which could be tuned by external magnetic fields.

Numerical Simulations. The optical response of the sample is calculated by using the finite-difference time-domain method in the Numerical FDTD Solutions software. In our simulations, the complex dielectric permittivity of the Ni film and Si disks as a function of frequency is taken from the standard dispersion data. The refractive index of the glass substrate is set constant 1.45 over the whole spectral range. In order to achieve the transmittance spectrum of the sample, we model an array of silicon disks covered with a 5 nm-thick Ni film and placed on a semi-infinite glass substrate, and illuminate it at the normal incidence by a plane wave, polarized perpendicularly to the magnetization. The calculated transmission spectrum is normalized to the power of the plane wave source. We use the periodic boundary conditions along both x and y axis and perfectly matched layers (PML) on the top and bottom of the unit cell to prevent parasitic interference. In order to obtain the magneto-optical response spectrum, the nondiagonal complex permittivity tensor of the Ni film is taken into account. Unitary transformation makes a permittivity tensor diagonalized with eigenvalues, and the Cartesian electric field components are converted into circularly polarized light according to the permittivity tensor. The MO response is defined as following: $2\frac{(T(H)−T(0))/T(0)}{T}$, where $T$ is the transmittance and $H$ is the external magnetic field.

■ ASSOCIATED CONTENT

Supporting Information

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Additional experimental data and numerical method description (PDF).

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